Influence of skin effect on the effective shielding effectiveness of composite materials

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Composite materials are increasingly used to contribute to structure lightening in electromagnetic shielding applications. The interactions between electromagnetic waves and composite materials are highly dependent on their microstructure. This gives rise to challenging modelling issues. Considering details of the microstructure would involve an excessive number of unknowns with standard numerical tools for structural analysis. Homogenisation methods—such as Maxwell-Garnett model—are a possibility to overcome this problem. The equivalent homogeneous medium obtained with such methods can be introduced into numerical tools to model full shielding enclosures. A homogenisation model has been recently proposed to obtain the equivalent homogeneous properties of composite materials subjected to electromagnetic waves. It relies on the introduction of a length parameter into classical non dimensional semi-analytical homogenisation methods—also known as mean field approaches. The model is applicable at microwave frequencies as long as the induced currents in the fibres (or inclusions) of the composite materials remain weak. This paper proposes an extension of the approach to include skin effect in the homogenisation method. This is done by considering Joule losses within the fibres of the composite. This extension significantly broadens the frequency range covered by the model. The results show that the optimization of composite shielding properties relies on a subtle compromise between internal reflections and Joule losses. © 2014 AIP Publishing LLC.

I. INTRODUCTION

Electromagnetic shielding enclosures contribute to electromagnetic compatibility. They are often made of metallic alloys. However, in the effort of reducing the weight of aircraft, the aerospace industry is considering the use of composite materials for shielding applications. Composites made of carbon fibres surrounded by epoxy resin are of particular interest.1 Experimental studies have been performed on such composites but accurate modelling tools are still lacking for the optimal design of these new generation enclosures. Numerical methods such as the finite element method can be used.2–8 These techniques are often performed on a periodic cell of the microstructure. This numerical homogenisation can be used to predict accurately the effective properties of the periodic cell. These effective properties can then be introduced in a structural analysis tool. If a large variety of electromagnetic loadings have to be considered, or if several microstructures have to be tested, these numerical methods can involve a significant computation cost, which can make them unsuitable, particularly in an engineering pre-design process.

An alternative approach is the use of semi-analytical homogenisation tools. Many homogenisation methods have been developed to model composite materials.9–15 Although they sometimes make use of empirical parameters, they appear to be a very efficient and fast modelling tool to obtain a first estimate of the effective properties of heterogeneous materials. Maxwell-Garnett model (MGM) is frequently used to model composite materials submitted to electromagnetic waves, but it is limited to low frequencies. A Dynamic Homogenisation Model (DHM), providing the effective properties at higher frequencies, has been recently proposed.16 An example of comparison between these two models and a standard Finite Element Model (FEM) is shown in Figure 1.

The figure is used here as an illustration of the motivations of this work, the simulation conditions are detailed in Sec. IV. If the FEM result is used as the reference, it can be seen that example that MGM is only valid up to 2 GHz. The DHM increases the validity range up to 10 GHz, but then moves away from the FEM solution. These discrepancies at high frequency can be explained by the appearance of eddy currents in the conductive phase of the composite. The small inserts in Figure 1 show the electric field intensity in the cross-section of a fibre at three different frequencies (1, 10, and 60 GHz). MGM gets inaccurate as soon as a rotating electric field appears within the fibre. This can be understood by recalling that MGM is equivalent to Hashin and Shtrikman (HS) estimate, obtained from a variational principle.17 HS estimate is known to be exact for a peculiar microstructure known as Hashin microstructure10 for which the fields are uniform in one of the phases. It then seems natural that the HS model diverges from the exact solution as the inhomogeneity of fields increases. The validity range of the DHM is larger but the model also gets inaccurate when the skin-effect gets significant.

The skin effect in the inclusions appears to be a key parameter for the homogenisation methods at high frequency. The present paper proposes to account for this effect by a

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A. Effective permittivity

For biphasic composites with isotropic constituents, if the HIM is chosen isotropic, the resolution of the two-inclusion problem provides the macroscopic complex permittivity $\varepsilon_\infty^*$ of the EHM in direction $\mathbf{u}$:

$$\varepsilon_\infty^* = \frac{f_1}{f_2} \frac{\varepsilon_1^* \varepsilon_\infty + N_\omega (\varepsilon_1^* - \varepsilon_\infty)}{\varepsilon_\infty + N_\omega (\varepsilon_1^* - \varepsilon_\infty)} + \frac{f_2}{f_1} \frac{\varepsilon_2^* \varepsilon_\infty + N_\omega (\varepsilon_2^* - \varepsilon_\infty)}{\varepsilon_\infty + N_\omega (\varepsilon_2^* - \varepsilon_\infty)}, \quad (1)$$

where $\varepsilon_\infty$ is the complex permittivity of the HIM and $N_\omega$ is the depolarisation factor along direction $\mathbf{u}$. For microstructures with infinitely long circular fibres, as studied in this paper, the depolarisation factors are $N_\omega = N_v = 1/2$ and $N_\omega = 0$. Equation (1) brings up the volume fraction $f_i$ and the complex permittivity $\varepsilon_i^*$ of phase $i$ (Phase 1: matrix and Phase 2: fibres). The complex permittivities are written

$$\varepsilon_i^* = \varepsilon_i - j \frac{\sigma_i}{\omega} \quad (2)$$

with $\omega$ the angular frequency, and $\varepsilon_i$ and $\sigma_i$ the electric permittivity and conductivity of the medium $i$. 

B. Homogeneous infinite medium

Particular choices of the HIM allow retrieving the results of classical homogenisation models. For instance, MGM can be obtained by choosing the matrix as the HIM ($\varepsilon_\infty = \varepsilon_1$). To extend the quasistatic approach to electromagnetic waves and form the DHM, the infinite medium is chosen as

$$\varepsilon_\infty = \varepsilon_1^* + \varepsilon_2^* \left( \frac{\gamma}{\lambda} \right)^2, \quad (3)$$

where $\lambda$ is the wavelength in the EHM and $\gamma$ is the characteristic size of the microstructure. This characteristic size is a function of the considered microstructure. In the 2D case of a square array of conductive discs, $\gamma$ is the diameter of the discs. For more complex microstructures, $\gamma$ can be determined by using 2-point probability functions. This extension was proposed to address the absence of any length parameter in MGM. The ratio $\gamma/\lambda$ is indeed a natural parameter to compare the characteristic length of the microstructure with the characteristic length of the electromagnetic wave. This size effect also depends on the fibre properties with higher effect in highly conductive materials. This is why $\varepsilon_2^*$ has been introduced in the definition of the infinite medium. Finally, $\varepsilon_1^*$ is introduced so that, at low frequencies (high $\lambda$), the model reduces to MGM (the second term in $\varepsilon_\infty^*$ (3) vanishes).

As shown in Figure 1, the DHM overestimates the Shielding Effectiveness (SE). This is attributed to an overestimation of the effective conductivity of the composite material, leading to a higher attenuation of the electromagnetic wave. Such limitations have been overcome in previous studies by introducing a renormalised conductivity accounting for the Joule losses in the material. In the present case, we choose to introduce the Joule losses in the definition of the HIM, using an equivalent losses conductivity (ELC) $\sigma_\infty$

$$\varepsilon_\infty^* = \varepsilon_1^* - j \frac{\sigma_\infty}{\omega}, \quad (4)$$
As displayed on Figure 2, the ELC is computed so that the energy dissipated by a fibre is deported in the inclusion problem to the infinite medium with conductivity $\sigma_\infty$.

The method is applied to 2D microstructures in the following. The extension to 3D microstructures requires more complex calculations for the Joule losses and is part of a work in progress.

III. JOULE LOSSES EQUIVALENCE

In order to define the proposed equivalence, the Joule losses in the real material are first computed analytically. The Joule losses in an infinite medium are then determined. Finally, the equivalence between these two energies defines the ELC.

A. Evaluation of Joule losses in the real material

1. Modelling configuration

The microstructure studied in this paper (Figure 4) is constituted of long parallel fibres (permittivity $\epsilon_s$, conductivity $\sigma_s$, and permeability $\mu_s$). To determine the energy lost in a fibre, the inner electromagnetic fields have to be computed. For the sake of simplicity, we use the semi-analytical expression computed for one fibre of the microstructure (Figure 3) surrounded by an homogeneous medium with properties ($\epsilon_i$, $\mu_i$). In the case of non-circular sections for the fibres, the diameter of an equivalent circular fibre, leading to the same Joule losses as the actual fibre, could be defined. Such equivalence can be defined analytically for elliptic sections but they cannot be solution of the differential Eq. (6). Since Bessel functions of the second kind are singular at the origin, they cannot be solution of the equation. The axial component of the internal magnetic field can finally be written

$$H_s(r, \omega, t) = H_0 \frac{\mathcal{J}_0(rk_2)}{\mathcal{J}_0(Rk_2)} e^{i\omega t},$$

where $\mathcal{J}_z(x)$ is the Bessel function of the first kind of order $z$. Since the tangential component of the magnetic field is equal on both sides of the interface ($r=R$)

$$H_0 = \sqrt{\frac{\epsilon_s}{\mu_s} E_0},$$

$H_0$ depends on the properties of the surrounding medium ($\epsilon_s$, $\mu_s$).

2. Inner magnetic field

From Maxwell equations and the axisymmetry of the problem, the differential equation governing the magnetic field is

$$r^2 \frac{\partial^2 H_z(r, \omega, t)}{\partial r^2} + r \frac{\partial H_z(r, \omega, t)}{\partial r} + (rk_2)^2 H_z(r, \omega, t) = 0$$

with $k_2 = \sqrt{\epsilon_s\mu_s\omega^2 - j\omega\sigma_s}$ the propagation constant in Phase 2. Bessel functions of the first and second kind of order zero are solutions of the differential Eq. (6). Since Bessel functions of the second kind are singular at the origin, they cannot be solution of the equation. The axial component of the internal magnetic field can finally be written

3. Inner electric field

The internal rotating electric field $E_i$ induced by the magnetic field (7) is obtained from Maxwell-Ampère equation

$$E_i(r, \omega, t) = \frac{k_2}{\sigma_2 + j\omega\epsilon_2} \frac{\mathcal{J}_s(rk_2)}{\mathcal{J}_0(Rk_2)} e^{i\omega t}.$$
5. Joule losses

To take into account the influence of the whole microstructure, the surrounding medium ($\epsilon_s$, $\mu_s$) is chosen as the MGM effective medium ($\epsilon_s = \frac{1+j\epsilon}{\rho_s}$; $\epsilon_1$) (see Appendix A) and $\mu_s = \mu_1$. The energy $W_{\text{Fibre}}^J$ lost in a fibre by Joule effect over a period (see Appendix B) is given by

$$W_{\text{Fibre}}^J = 2\pi^2 h_1 \frac{1 + f_2}{f_1} E_0^2 f_1^R \left| \frac{J_1(rk_2)}{J_0(Rk_2)} \right|^2 dr,$$

where $h$ is the height of the fibres. Note that the dependence of this energy to the fibre conductivity $\sigma_2$ is hidden in the expression of $k_2$. The surrounding medium being non-conductive, this expression of losses corresponds to the whole Joule losses in the unit cell (association of one fibre with volume fraction $f_2$ and surrounding medium).

B. Evaluation of Joule losses in the infinite medium

The local energy lost by an infinite isotropic homogeneous medium where there is an electric field $E = E_0e^{j\omega t}$ is equal to

$$w_J^\infty = \int_0^T J_0(r, \omega, t) E_0(r, \omega, t) dt = \frac{\pi}{\omega} \sigma_\infty E_0^2,$$

The energy lost by a volume of infinite medium equal to the volume ($\pi R^2 h_1 f_2$) of the unit cell considered in Sec. III A is

$$W_J^\infty = \frac{\pi R^2 h_1}{f_2} w_J^\infty = \frac{\pi^2 R^2 h_1}{f_2 \omega} \sigma_\infty E_0^2.$$

C. Determination of the ELC

It is found that the best choice for the conductivity of the infinite homogeneous medium in the inclusion problem is obtained by deporting integrally the losses in the fibres to the infinite medium. This transfer is expressed by the equality

$$W_J^\infty / f_1 = W_{\text{Fibre}}^J / f_2,$$

$W_J^\infty / f_1$ and $W_{\text{Fibre}}^J / f_2$ are the losses in the HIM and in the fibres, respectively, scaled to their proportion in the inclusion problem. Introducing (11) and (13) in this equality leads to the following definition for the ELC

$$\sigma_\infty = \frac{2}{R^2} \left( 1 + f_2 \right) \epsilon_1 \int_0^R \left| \frac{J_1(rk_2)}{J_0(Rk_2)} \right|^2 dr.$$ (15)

IV. MODELLING RESULTS

To evaluate the accuracy of the EDHM, the SE of a composite material sheet is compared to the SE of a homogeneous sheet with the EHM properties. The SE is classically defined by

$$SE = 20 \log_{10} \frac{E_I}{E_T},$$

where $E_I$ and $E_T$ are the norms of the incident and transmitted electric fields. The SE of the sheet made of the EHM is calculated analytically and the SE of the corresponding heterogeneous sheet is computed by FEM on a mesh of the complete microstructure as shown in Figure 4. We focus in the following on the case where the electric field is perpendicular to the fibres. An electric field parallel to the fibres does not allow a discriminating comparison since the Wiener estimate is generally sufficient to determine the SE in that particular case.\(^2\)

A. Analytical solution for the shielding effectiveness

When the shielding sheet is illuminated by a perpendicular plane wave, the analytical solution of (16) can be expressed as (see Ref. 16)

$$SE = SE_A + SE_B - SE_R,$$

where $SE_A$ is the attenuation of the wave caused by the absorption of the material

$$SE_A = 20 \log_{10} |e^{j\delta}|$$

with $k = \sqrt{\epsilon j \omega^2 \mu j \sigma \omega}$, $\epsilon$ and $\sigma$ are the permittivity and conductivity of the material in the direction of the incident electric field $E_I$ and $\mu$ the magnetic permeability in the direction of the incident magnetic field $H_I$. The second attenuation is caused by reflection

$$SE_R = 20 \log_{10} |p|,$$

where $p = \frac{n^2}{\mu} (\frac{n^2}{\mu} + 1)^2$ is the transmission coefficient depending on $n = k / k_0$, the refractive index of the medium. The third contribution corresponds to multiple reflections inside the material

$$SE_B = 20 \log_{10} |1 - q^2 e^{-2j\delta}|,$$

where $q = (\frac{n^2}{\mu} - 1) / (\frac{n^2}{\mu} + 1)$ is the reflection coefficient.

B. Finite element calculations

Due to computational resources limitations, numerical calculations have been processed on a small domain having a height of 0.1 mm (Figure 4) using COMSOL Multiphysics.\(^8\) To simulate infinity, this domain is bounded by two perfectly matched layers (PMLs). Neumann boundary conditions are applied on top and bottom of the region modelled to simulate the periodicity of the infinite sheet. The sheet is 6 mm thick and filled with a single row of 60 cylindrical fibres (circular cross section), regularly spaced. These fibres have a diameter of 50 $\mu$m leading to a fibre volume fraction $f_2 = 19.63\%$. No contrast in permeability is considered ($\mu_1 = \mu_2$). The incident electromagnetic wave is a
perpendicular plane wave propagating along the $x$ axis, electrically polarised along $y$. For a given frequency, the computation time with FEM is about 45 s on a standard workstation. The homogenisation model only takes some milliseconds on the same workstation.

As already shown in Figure 1, the DHM, with HIM defined by (3), increases the frequency range of MGM but gets inaccurate when the skin effect becomes significant. The EDHM, with HIM defined by (4), still provides an accurate estimate at high frequencies when compared to FEM results. The case studied here gives a null SE at low frequencies because the matrix is not conductive. Results for MGM and DHM with a conductive matrix can be found in Ref. 20.

The EDHM can be used with various constituent properties. An example is shown in Figure 5. The microstructure is the same as in Figure 1, and the material properties are those used by De Rosa et al. for the design of a composite made of short carbon fibres embedded in an epoxy resin.

All three models reproduce the oscillations of the SE as a function of the frequency. These oscillations are due to the multiple reflections of the wave on the material-air interface. These reflections depend on the thickness of the sheet and on the complex permittivity of the material. The SE tends to increase with frequency because the Joule losses increase with frequency. This effect is linked to the role of the effective conductivity of the composite, which is underestimated by the MGM, and overestimated by the DHM. The predictions of the EDHM are in good agreement with the FEM computations.

V. DISCUSSION

In order to assess the validity of the proposed EDHM, calculations have been performed under various configurations in terms of fibre diameter, fibre conductivity, fibre volume fraction, and matrix permittivity.

Figure 6 shows the difference between the homogenisation results (MGM, DHM, and EDHM) and the FEM results for the computations performed in this paper. Each marker represents the error with respect to the FEM computation calculated for a given set of parameters (frequency, fibre density, fibre diameter, fibre conductivity, and matrix permittivity). The results are plotted as a function of the ratio between the diameter of the fibres and the skin depth in a fibre ($\delta_2 = \sqrt{2/\mu_2\sigma_2\omega}$). The discrepancy between MGM and FEM results increases as soon as the skin depth of the fibre becomes equal to the diameter of the fibres $\delta_2$. As shown in Ref. 16, the DHM extends MGM frequency range up to high ratios between the diameter and the skin depth of the fibres. It is clear from Figure 6 that the proposed approach significantly extends the frequency validity range of semi-analytical homogenisation method for the prediction of the SE of composite materials.

Figure 7 shows the results obtained with the EDHM and the FEM for different fibre diameters, the other properties being kept constant. The agreement is very good between the two methods although slight differences appear for the diameter of 100 $\mu$m. These differences can be attributed to the fact that for high fibre diameters, the problem computed with FEM is not really a Representative Volume Element. If the finite element calculation was done on a larger surface including more fibres, it is expected that the two results would be closer. Another cause could be an insufficient precision in the definition of the permittivity of the HIM. Indeed it can be shown that a better match could be obtained between EDHM and FEM with a slight change in the real parameter.
part of the HIM properties (4). This point highlights the crucial role played by the choice of the infinite medium in homogenisation approaches based on inclusion problems. Other definitions of the HIM properties can be investigated but the proposed choice (4) is the best obtained so far in terms of accuracy and versatility.

Another interesting observation is that the quality of the SE is not a monotonic function of the fibre diameters. If a high diameter is chosen, the skin effect appears at lower frequencies, inducing a better SE. But on the other hand at high frequency, the inner part of the fibres is free from currents since the skin effect tends to move the rotating currents towards the edge of the fibres. The area with no current is then larger with a large-fibre microstructure, creating relatively low Joule losses, and as a consequence decreasing the SE compared to small-diameter microstructures. The choice of an optimal diameter for the fibres of shielding composites is then a compromise between these two competitive processes.

Figure 8 shows the results obtained with the EDHM and the FEM for different fibre electric conductivities, the other properties being kept constant. The agreement between FEM and EDHM is very satisfying. At high volume fractions, small more significant and high conductivity fibres create less Joule losses, which is then unfavourable for SE. This effect is included in the EDHM through the choice of the infinite medium (15) that is less conductive when Joule losses are lower. Again, the choice of the optimal properties for the fibres is a compromise between these two competitive processes.

Figure 9 shows the results obtained with the EDHM and the FEM for different fibre volume fractions, the other properties being kept constant. The agreement between FEM and EDHM is very satisfying. At high volume fractions, small
differences appear. They can be linked to the choice of Maxwell-Garnett approach for the choice of the infinite medium. This kind of approach is known to be valid only for small volume fractions. At high volume fractions, a self-consistent approach would probably give a better agreement. The implementation of a self-consistent approach will be proposed in a further communication. As expected, Figure 9 shows that the higher the proportion of fibres, the better the SE.

Figure 10 shows the results obtained with the EDHM and the FEM for different matrix permittivities, the other properties being kept constant. The agreement between FEM and EDHM is very satisfying. The role of the matrix permittivity on the effective properties is rather complex. It affects both the effective permittivity and the effective conductivity. A high matrix permittivity increases the conductivity of the infinite medium (15), tending to increase the SE. On the other hand, the multiple reflections in the sheet heavily depend on the permittivity and on the frequency (20). A high matrix permittivity tends to increase the level of oscillations of SE as a function of frequency. This effect is well reproduced in Figure 10.

VI. CONCLUSION

In this paper, a semi-analytical homogenisation approach for the definition of the shielding effectiveness of composite materials is proposed. Contrarily to previous approaches, it is shown to be valid for electromagnetic waves at high frequencies—up to tens of GHz in the examples provided. The model is based on the resolution of basic inclusion problems. An appropriate choice of the infinite medium surrounding these inclusions is key to these approaches. A choice accounting for the Joule losses in the fibres is used for the proposed model. The results show very good agreement with finite element computations performed independently—and at much higher computational cost. An interesting outcome of the study is that contrarily to shielding sheets made of homogeneous materials where a conductivity as high as possible is sought in order to maximise the reflection effects, the choice of an optimal composite material for shielding effectiveness is a more subtle process. A compromise must be found between reflection effects and Joule losses within the fibres in order to maximise the overall shielding effectiveness. The proposed homogenisation method is a convenient tool to reach such an optimum, and can be readily used for pre-design process or for an implementation into standard numerical tools of full devices with complex geometries.

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APPENDIX A: MAXWELL-GARNETT ESTIMATE

The Maxwell-Garnett estimate is intensively used in this paper. In 2 dimensions, Maxwell-Garnett model is usually written

\[ \tilde{\varepsilon}^*_{MG} = \varepsilon_1 + 2f_2\varepsilon_1 \frac{(\varepsilon_2 - \varepsilon_1) - j\frac{\sigma_2}{\omega}}{(\varepsilon_1(1 + f_2) + f_1\varepsilon_2) - f_1f_2\frac{\sigma_2}{\omega}}. \]  

(A1)

By considering (A4) and (A3), (A1) can be written

\[ \tilde{\varepsilon}^*_{MG} = \varepsilon_1 + 2f_2\varepsilon_1 \frac{(\varepsilon_2 - \varepsilon_1) - j\frac{\sigma_2}{\omega}}{(\varepsilon_1(1 + f_2) + f_1\varepsilon_2) - f_1f_2\frac{\sigma_2}{\omega}}. \]  

(A4)

In the particular case of this article, the conductivity of fibres (index 2) is very high. This leads to \( \sigma_2/\omega \gg \varepsilon_1 \) and \( \varepsilon_2 \). In such conditions, the previous expression for the Maxwell-Garnett model can be simplified. The real part is given by

\[ \tilde{\varepsilon}_{MG} \approx 1 + \frac{\varepsilon_2}{f_1}. \]  

(A5)

and the imaginary part by

\[ \frac{\tilde{\sigma}_{MG}}{\omega} \approx \omega Af_2\varepsilon_1^2\frac{\sigma_2}{f_1\tilde{\varepsilon}_{MG}}. \]  

(A6)

The conductivity of the Maxwell-Garnett medium can then be neglected, and its permittivity can be written according to (A5).

APPENDIX B: JOULE LOSSES IN A FIBRE

The local energy lost by Joule effect \( w_{Fibre}^{Joule} \) in a fibre (\( \varepsilon_2 \) and \( \mu_2 \)) surrounded by an homogeneous medium (\( \varepsilon_s \) and \( \mu_s \)) over a period \( (T = 2\pi/\omega) \) is computed by
Equation (B4) can be simplified by considering the fibres highly conductive \((\sigma_2/\omega \gg \epsilon_2)\), and by choosing the Maxwell-Garnett estimate for the surrounding medium \((\epsilon_s = \frac{1 - f_2}{f_1} \epsilon_1 \text{ and } \mu_s = \mu_1 = \mu_2)\)

\[
 w_{Fibre}^{\text{Fibre}} \approx \epsilon_s \frac{1 + f_2}{f_1} \frac{\mu_2}{\mu_0} \left| \frac{J_1(rk_2)}{J_0(Rk_2)} \right|^2.
\]  

(B5)

The total losses \(W_j^{\text{Fibre}}\) are computed by integrating the local energy over the radius since the integrated fields only depends on \(r\)

\[
 W_j^{\text{Fibre}} = \pi \int_0^R 2 \pi r w_{Fibre}^{\text{Fibre}} dr,
\]  

(B6)

\[
 W_j^{\text{Fibre}} \approx 2 \pi^2 h \epsilon_1 \frac{1 + f_2}{f_1} E_0^2 \int_0^R \left| \frac{J_1(rk_2)}{J_0(Rk_2)} \right|^2 dr,
\]  

(B7)

where \(h\) is the height of the fibres.